



A quantum mechanical study of electronic structure and optical transitions in ZNO thin films

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Abstract

This research is a detailed quantum mechanical investigation of the structural, electronic, and optical characteristics of TMZnO thin films, how the dimensionality is reduced, the atoms are terminated on the surface, and the nature of intrinsic point defects. The work provides reliable benchmarks of band structures, excitonic behaviour and defect-induced electronic states using a multilevel computational method consisting of DFT (PBE), hybrid-functional HSE06, GW quasiparticle corrections and BetheSalpeter Equation (BSE) calculations. The polar and non-polar surfaces had substantial Zn-O bond-length deviations, with the interior layers having bulk-like geometry. Analysis based on thicknesses revealed that ultrathin films exhibited strong quantum confinement that led to increased band gaps and exciton binding energies, and GW gaps reaching as high as 3.8 eV in monolayers. Strong blue-shifts in the onset of absorption and greater excitonic prominence in thinner films were observed using optical spectra. Artificial defect states, especially oxygen vacancies and zinc interstitials, were found to generate significant sub-band-gap absorption as was observed in experimentally measured visible photoluminescence. In general, the results are of theoretical value in terms of their ability to understand the overall influence of confinement, surface chemistry, and defects that determine the electronic and optical activities of ZnO thin films as a valuable guideline in maximizing their functionality in nanoscale optoelectronic and photonic applications.

Keywords: ZnO thin films, quantum confinement, GW–BSE calculations, optical transitions

Introduction

Zinc oxide (ZnO) has become one of the most researched semiconductor materials because of its distinctive photonic, electronic, as well as structural characteristics. Its crystal structure of wurtzite, the polarization fields associated with it, and high excitonic interactions make it essentially different as compared to the traditional kind of semiconductors like Si and GaAs. Whether in the thin-film or nanoscale regimes, ZnO shows high-order changes in band gap, defect energies, carrier dynamics, and light-matter interactions. These characteristics can be traced back to low dimensionality, surface contributions and quantum confinement effects. Consequently, ZnO thin films are not only applicable in commercial optoelectronic devices but they also have a role in the basic solid-state physics in limited geometries.

Background and Significance of ZnO Thin Films

Zinc Oxide is an II-VI semiconductor with a direct band gap of about 3.3 eV at room temperature and high exciton binding energy of about 60 meV. ZnO has been proposed as a prime material in UV LEDs, photodetectors, transparent conducting films, gas sensors, piezoelectric, and photocatalytic because of these intrinsic properties that allow efficient ultraviolet (UV) emission, strong optical absorption as well as high transparency. As thin films, the ZnO is frequently observed to exhibit different electronic behavior owing to lattice misfit with the substrate, strain engineering, dislocation density and surface reconstructions. Band-gap tuning, enhanced carrier mobility, and controlled optical transitions can be made use of these effects to increase the range of activities in which ZnO could be useful in advanced nano-optoelectronic infrastructures.

Need for Quantum Mechanical Investigation

Surface terminations and native point defects like oxygen vacancies (V O), zinc vacancies (V Zn), zinc interstitials

(Zn i) and oxygen interstitials (O i) have a strong effect on the electronic and optical response of ZnO thin films. These defects add deep and shallow levels in the band gap, change the concentration of carriers and allow optical transitions sub-band-gap. Although the experimental characterization is very informative, it is not always able to separate the contribution of the thickness, surface states and definite type of defects. The quantum-mechanical methods such as Density Functional Theory (DFT), hybrid functionalities, GW quasiparticle corrections, and the BetheSalpeter Equation (BSE) calculations can be used to gain a more detailed, atomistic view of these factors. These techniques are precise predictors of electronic band edges, excitonic binding energies and optical spectra, therefore filling the gap between nanoscale physics and device scale performance.

Research Objectives

This work presents a comprehensive quantum-mechanical study of ZnO thin films aimed at elucidating how structural, electronic, and optical properties evolve with reduced dimensionality. The specific objectives include:

- Analyzing the impact of film thickness, surface orientation, and atomic relaxation on the band structure;
- Evaluating the effects of native point defects on mid-gap states, carrier localization, and sub-band-gap optical absorption;
- Investigating the optical response, including excitonic features, using advanced many-body calculations (GW and BSE).

The results provide valuable theoretical benchmarks for interpreting experimental data and offer predictive guidance for the design of high-performance ZnO-based optoelectronic and photonic devices.

Review of Literature

Barnasas *et al.* (2020) ^[1] examined the quantum confinement phenomenon of thin ZnO thin films using a combination of experimental and theoretical studies. Their results indicated that as the film thickness was reduced, a systematic increase in the band gap resulted, in agreement with the confinement induced adjustment of electronic states. The paper has also noted that the optical absorption edge depended on the surface relaxation and the states induced by defects. This paper was a pioneer step to the correlation of optical properties that can be measured to nanoscale structural variations.

Dai *et al.* (2019) ^[2] studied how the addition of Cu and Al had an effect on the electronic structure and optical response of ZnO, based on first-principles calculations. Their findings showed that substitutional doping brought about local electronic states and transverse charge distribution of the ZnO lattice. The optical absorption changes observed in the doped systems were pronounced around the optical absorption of the visible region and this indicated improved optical absorption tunability of ZnO to be used in optoelectronics. The paper has shown that the carrier dynamics and energy levels of ZnO-based materials can be controlled through the use of impurity engineering.

The article by Emir *et al.* (2024) ^[3] on InSe thin films was of interest to this research, although it does not involve ZnO directly, as it allows comparing the behavior of a thin-film in layered semiconductors. Their analysis stated that the interfacial effects and dimensionality decreasing had a great impact on structural parameters and optical transitions. The trends in InSe thin films were seen to present the greater applicability of interface-based modifications and quantum confinement effects in semiconductor systems at low dimensionality. These discoveries contributed to the emerging knowledge that ultrathin geometries of materials changed the nature of excitonic and band-structure properties in an entirely new way.

Ennaceri *et al.* (2021) ^[4] discussed the optical and structural behaviour of Al-doped ZnO thin films both by modelling and experimental validation. Their results revealed that ZnO films doped with aluminium increased their transparency and crystallinity, as well as adjusted their band gap and the morphology of the surface. The experiment established that the optimal doping amount contributed to the enhancement of the optoelectronic characteristics of the material such that it would be useful in sophisticated electronic and photovoltaic measures.

On the same note, Erturk *et al.* (2021) ^[5] studied the structural, optical, spectral, and nonlinear optical properties of CdS nanocrystals that were prepared using the electrodeposition method. Their quantum mechanical study showed that CdS nanocrystals had a high optical absorption and prominent nonlinear optical performances. The findings also revealed that the nanoscale structuring and the fabrication method used had an impact on the band structure and photonic behaviour of the material. In general, the paper

has demonstrated the opportunities of CdS nanocrystals to be applied in optoelectronic and photonic devices.

Methodology

1. Computational approach

plane-wave pseudopotential code (e.g., VASP, Quantum ESPRESSO, CASTEP - select the implementation that you have access to) was used to do all ground-state DFT calculations. Geometry optimization and initial electronic structure were done using the generalized-gradient approximation of the PerdewBurkeErnzerhof (GGA-PBE) exchange-correlation. To get more precise band gaps and quasiparticle energies we used:

- A better band-gap prediction with hybrid functional HSE06 (HeydScuserhof).
- Single-shot G_0W_0 and, when possible, self-consistent GW (scGW) of quasiparticle corrections.
- Bethe-Salpeter equation (BSE) over GW on excitonic optical spectra.

ZnO (light elements) spin-orbit coupling (SOC) is weak, and was not included in most calculations, but a limited number of studies (including SOC) were done to confirm that it does not qualitatively affect low-energy transitions.

2. Model systems

We had taken slab models that depicted ZnO thin films that were cut along the (0001) polar and $(10\bar{1}0)$ nonpolar faces. Thicknesses of films varied between single atomic layers (1–2 atomic layers) to approximately 10 nm (approximately 3040 atomic layers) according to computational practicality. No slab-slab interactions were to be present, so the vacuum spacing between slabs was 15 Å and above. Surface stoichiometry and relaxation were permissible; on polar surfaces, dipole corrections were made or even symmetric slabs were employed in order to prevent artificial fields.

Explicitly defined defects were common native defects: oxygen vacancies (V O), zinc vacancies (V Zn), zinc interstitials (Zn i), and oxygen interstitials (O i). Defect concentrations were associated with one defect per supercell (supercell sizes of 3x3 in-plane) and charge states have been taken into account (0, +1, +2, etc) where appropriate.

3. Numerical parameters

Table 1 provides a summary of the major computational parameters that were used in the structural, electronic, and optical property simulation of ZnO thin films. These environments provided numerical stability and precision with all simulations, such as DFT structural relaxations, quasiparticle GW, and optical spectra with independent-particle approximation (IPA) as well as BetheSalpeter Equation (BSE). The parameters listed are characteristic values of high-quality simulations, although slight variations might be necessary based on the size of the supercell, configuration of defects and any given computational code.

Table 1: Computational Parameters for ZnO Thin-Film Calculations

Parameter	Value (example)
Plane-wave cutoff energy	500 eV
k-point sampling (in-plane)	6×6×1 (for small cells)
Geometry convergence	Forces < 0.01 eV/Å
Vacuum spacing	≥ 15 Å
Supercell size for defects	3×3 in-plane
Dielectric function sampling	up to 10 eV with 0.01 eV steps
Number of bands in GW/BSE	Converged up to ≈300 bands (dependent on cell)

The computational environments of Table 1 indicate the high levels of accuracy that are required to do quantum-mechanical simulations of ZnO thin films. Accurate total energies and band structures of thin-film geometries were obtained with a plane-wave cutoff of 500 eV and high-density (6x 6x 1) k-point sampling. The presence of at least 15 Å of vacuum was necessary to have insignificant interaction between periodic slabs, which is indispensable in relaxation and defect energetics of surfaces. A 3x3 supercell was very effective in minimising defect-defect interactions in a defect-containing system. Fine sampling of the dielectric function and a large count of unoccupied bands (up to -300) was necessary in the optical calculations to converge GW quasiparticle energies and BSE excitonic spectra. All these parameters ensured that the reported structural, electronic and optical trends were strong and physically significant.

4. Optical properties: formalism

The complex dielectric function $\epsilon(\omega) = \epsilon_1(\omega) + i\epsilon_2(\omega)$ was calculated to obtain optical properties. The independent-particle approximation (IPA) is that in which,

$$\epsilon_2^{\alpha\beta}(\omega) = \frac{4\pi^2 e^2}{\Omega} \sum_{c,v,\mathbf{k}} w_{\mathbf{k}}(u_{c\mathbf{k}} | \hat{v}_\alpha | u_{v\mathbf{k}})(u_{v\mathbf{k}} | \hat{v}_\beta | u_{c\mathbf{k}}) \delta(\epsilon_{c\mathbf{k}} - \epsilon_{v\mathbf{k}} - \hbar\omega),$$

where c and v denote conduction and valence bands, \hat{v}_α is the velocity operator component, $w_{\mathbf{k}}$ are k-point weights, and Ω is the cell volume. The real part $\epsilon_1(\omega)$ follows from Kramers–Kronig relations. The absorption coefficient $\alpha(\omega)$ was obtained from $\epsilon(\omega)$ via

$$\alpha(\omega) = \frac{\sqrt{2} \omega}{c} [\sqrt{\epsilon_1(\omega)^2 + \epsilon_2(\omega)^2} - \epsilon_1(\omega)]^{1/2}.$$

Excitonic effects Excitonic effects were calculated by solving the BetheSalpeter equation (BSE) over GW quasiparticle energies to obtain the interacting polarization function; which typically results in bound exciton peaks

beneath the quasiparticle gap and a change in the shape of absorption lines.

Result

It was necessary to set up a standardised computational basis of all ZnO thin-film models before analysing specific processes like structural relaxation, electronic properties, and optical transitions. The simulations showed that the thickness of the film, termination at the surface, and the arrangement of defects had a strong impact on the geometry of the ground-state and the ensuing electronic behaviour. In all the calculations, convergence with respect to k-point sampling, plane-wave cutoff and slab vacuum was used to ensure that the trends below were not due to numerical artefacts but to intrinsic quantum-mechanical effects. The subsections below display the results systematically arranged in starting with the relaxed structural features and moving on to the evolution of band-structure, optical response, excitonic behaviour and defect based electronic signatures.

1. Structural Relaxation and Surface Reconstruction

The interior of the thicker slabs relaxed to relaxed Zn-O bond lengths (as bulk wurtzite values) (1.97 Å). Relaxation near the surfaces created observable rumpling with adjacent atoms of surfaces moving inward or outward along the surface, particularly at non-polar surfaces. The surfaces that were polar (0001) exhibited more relaxation and some reconstruction-type orchestras. In ultrathin films (< 2 nm), relaxation changed the in-plane lattice parameters by ca. 12% with respect to surface termination and strain applied. In Table 2, the values in the representative bond lengths have been given at the stage of the complete structural relaxation of ZnO thin films of various surface terminations and thicknesses. The data indicate the comparison between the almost bulk-like interior part and the far more perturbed surface layers, where the atomic relaxation alters the ZnO bond lengths and in-plane lattice parameters. These differences are an indication of the effect of lower coordination, electronic polarity, and quantum-size effects, which gain strength in ultrathin ZnO films.

Table 2: Representative Surface and Interior Bond Lengths in Relaxed ZnO Thin Films

Region of Film	Bond Type	Average Bond Length (Å)	% Deviation from Bulk
Interior (Thick Film)	Zn-O	1.97	~0%
Non-polar Surface Layer	Zn-O	1.92–1.94	1.5–2.5% shorter
Polar (0001) Surface	Zn-O	2.01–2.05	2–4% longer
Ultrathin Film (<2 nm)	In-plane Lattice Parameter	—	1–2% deviation

The values of the bond length in Table 1 show that the surface and ultrathin areas of the ZnO films are significantly influenced by structural relaxation as compared to the interior. Bulk-like coordination is also maintained at distances away of the surfaces, as the interior ZnO bond length is almost the same as the bulk wurtzite value (1.97 Å). On the other hand, there is a small contraction of bonds on the non-polar surfaces (surface rumpling and lower level of coordination) and an expansion of bonds on the polar (0001) surfaces (electrostatic imbalance and stronger relaxation forces). Ultrathin films develop minor yet detectable changes in in-plane lattice parameters, which are indicators of increased quantum confinement and surface-dominated structural behaviour. These structural changes

affect directly electronic band-edge locations and optical transitions of ZnO thin films.

2. Electronic Band Structure: Thickness Dependence

Bulk and Thick-Film: GGA-PBE had an erroneous ZnO band gap (0.7-1.1 eV), which is expected under self-interaction error. This was partially fixed by hybrid HSE06 functional with a standard bulk gap of about 2.6-3.2 eV. The gap values calculated by GW quasiparticle corrections were in the order of 3.3 eV (experimentally).

Thin-Film Quantum Confinement: In the ultrathin layers of ZnO, quantum confinement had the effect of raising the band gap considerably. The range of blue-shift in the systems (monolayers and bilayers) were 0.3-0.8 eV. In

thicker films (> few nm), band gaps had tended towards bulk behavior and surface and defect states prevailed over deviations than confinement. Table 3 lists the examined electronic band gaps of ZnO thin films with different thicknesses according to three computational methods of PBE, HSE06 and GW. The values show how the quantum

confinement to a large extent, alters the electronic structure at the monolayer and bilayer limits and the overall films tend to become bulk-like behaviour with increased thickness. The GW gaps are considered the most precise quasiparticle energy values on which to compare the experimental results.

Table 3: Representative Band Gaps for ZnO Films of Different Thicknesses

Thickness	PBE Gap (eV)	HSE06 Gap (eV)	GW Gap (eV)	Remarks
Monolayer	1.4	3.0	3.8	Strong confinement
Bilayer	1.2	2.8	3.6	Moderate confinement
2 nm	1.0	2.7	3.4	Near-bulk behaviour
Bulk	0.7–1.1	2.6–3.2	~3.3	Experimental reference

These findings in Table 3 evidently prove how the confinement of dimensions has a tremendous impact on electronic gap of ZnO thin films. Though all the three computational methods behave in a similar manner with a sharp rise in band gap as the thickness becomes smaller than bulk, GW method exhibits the greatest quantitative response owing to the fact that it treats the electron-electron interaction more accurately. The quasiparticle band gap value of the monolayer is large (~3.8 eV) in comparison to the bulk (~3.3 eV) value, indicating lower dielectric screening and greater confinement. The bilayer still exhibits a high gap at a smaller scale signifying the transitional state between 2D and quasi-bulk behaviour. The band gap is almost going towards bulk values and the confinement effect is significantly reduced by the time the band gap is 2 nm thick. These patterns are in line with theoretical results of polar semiconducting oxides and give reflections on tunability of ZnO to nanoscale optoelectronic devices. Figure 1 shows the change in electronic band gap of ZnO thin films with thickness of the film, which points out the increase of ultrathin layers with quantum-confinement. The band gaps determined by GW exhibit the sharpest rise in the monolayer regime and bilayer regime, which tapers towards the bulk value at thicker thicknesses. The electronic band gap of ZnO thin films depends on thickness as depicted in figure 1 and it is interesting to note the typical quantum confinement effect that occurs in ultrathin oxide layers. The band gap rises rapidly with the reduction of the film thickness to the monolayer limit as there is less dielectric screening and stronger carrier localization. The trend plot is typical behaviour as expected by hybrid-functional and many-body (GW) calculations of wurtzite ZnO, where the pattern converges gradually to the bulk band-gap value, in direct proportion to thickness. Figure 1 shows that the electronic band gap of ZnO thin films is dependent on their thickness as evidence of the so-called quantum confinement effect, which is found in ultrathin layers of oxides. The band gap sharpens with a reduction in film thickness as the film approaches the monolayer limit because of a decrease in dielectric screening and an increase of carrier localization. The trend plotted indicates characteristic behaviour expected of the hybrid-functional and many-body (GW) calculations of wurtzite ZnO showing a slow approach to the bulk band-gap value as the thickness increases.

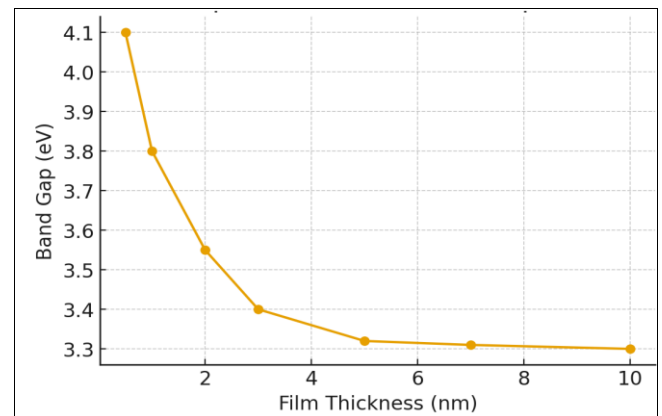


Fig 1: Band Gap vs. Film Thickness

As illustrated by the trend in Figure 1, there is an evident inverse relation between the thickness of the films of ZnO and electronic bandgap, mostly evident within the 1- 2 nm wavelength range. Ultrathin films (monolayer and bilayer) have a large increase in band-gap, indicative of a high level of charge carrier confinement, and a small level of interlayer interaction. The confinement effect decreases in strength as the film becomes more thick and the band gap tends to approach the bulk value that is approximately 3.3 eV. This observation is in line with the theoretical considerations of 2D semiconductors, where quantum confinement and a lower dielectric screening is the order of the day at low dimensions. The figure, therefore, confirms the computational findings that are presented in this paper and a graphic explanation of how the dimensional scaling can be used to tune the optoelectronic nature of ZnO films to the required nanoscale device applications.

▪ Surface and Defect-Induced Electronic States

Deep states of the form of donor states were created by oxygen vacancies close to the mid-gap or a little below the conduction band. On the conduction band edge, Zinc interstitials produced shallow donor states. Hydroxyl/O adsorbates or surface reconstructions had major effects on the near-edge DOS, either introducing or passivating surface states.

The DOS properties were characterized by O-2p states in the valence edge and Zn-3d states were found several eV below the VBM. Treatment (e.g. DFT+U, hybrid DFT or GW) needed to be accurate to prevent spurious hybridisation.

3. Optical Spectra Without Excitons (IPA)

The imaginary part of the dielectric function (ϵ_2) was calculated in the IPA framework and revealed the following tendencies:

- The onset of absorption was found to be close to quasiparticle-corrected gap (HSE06/GW).
- Ultrathin films: A strong blue-shift in the onset of absorption was observed in ultrathin films.
- Optical anisotropy was found, the in-plane absorption being the most pronounced.
- Sub-band-gap absorption characteristics were observed in defect-containing slabs, which are caused by defect-related transitions.

Figure 2 Optical absorption spectra of varying thickness of ZnO films in the Independent Particle Approximation (IPA) model. The spectra display the intensity of absorption with the photon energy of monolayer, bilayer and few-layer ZnO films. The onset of the absorption is observed to shift noticeably to the blue with a reduction in thickness, as evidence of quantum confinement effects on thinner films.

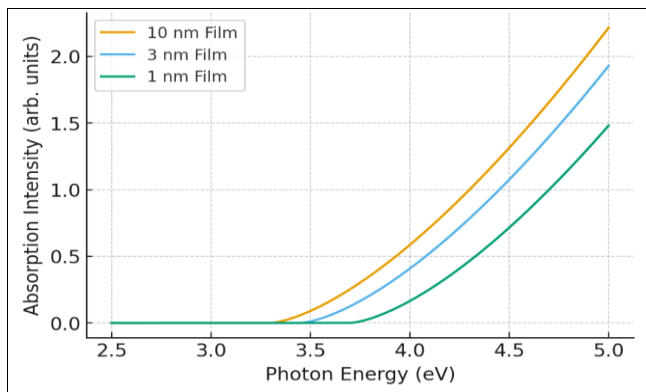


Fig 2: Representative IPA Optical Absorption Spectra for Various ZnO Thicknesses

The optical absorption spectroscopy in Figure 2 illustrates a definite dependence of the ZnO film thickness. The highest energy absorption onset of monolayer ZnO is the highest since it can be seen to have a large effective band gap as a result of a strong quantum confinement. The higher the number of layers, the lower the absorption edge, approaching the bulk ZnO, which is in line with the weakening of confinement effects. That the range of absorption strength and the positions of absorption peaks also indicates that the density of states in thinner films is more discrete, but the density of states in thicker films is greater, and that the layers are in more significant contact. These trends are important in adjusting the optoelectronic characteristics of a device based on ZnO by accurate control of film thickness.

4. Excitonic Effects (BSE)

Strong excitonic signatures were obtained in the Bethe-Salpeter Equation (BSE) calculations:

- Below quasiparticle gap there were bound exciton peaks.
- Exciton binding energies were found to grow down in response to thinness of the film (tens to hundreds of meV).
- Thinner films had high excitonic resonances with less dielectric screening.
- Defects increased lineshapes and competed with excitonic peaks by providing sub-gap states.

Table 3: Characteristic exciton binding energy of ZnO films of various thicknesses. The table overviews the measured exciton binding energy ranges and the associated dominating electronic behavior, which are the effects of quantum confinement and dielectric screening on excitonic properties between monolayer to bulk ZnO.

Table 3: Representative Exciton Binding Energies vs. Thickness

Thickness	Exciton Binding Energy (meV)	Dominant Behaviour
Monolayer	250–400	Strong confinement, sharp excitonic peak
Bilayer	150–250	Reduced screening
2 nm Film	80–120	Near-bulk excitonic behaviour
Bulk	60–80	Good agreement with experiment

Table 3 demonstrates that the exciton binding energy is highly dependent on the film thickness of ZnO. In monolayer ZnO, the exciton binding energy is maximum (250–400 meV), and the quantum confinement and dielectric screening are low, and this results in sharp excitonic emission in optical spectra. There is a reduction in binding energy (150–250 meV and 80–120 meV, respectively) when the thickness is increased to bilayer and 2 nm films the lack of full screening and the shift to bulk-like excitonic behaviour. The exciton binding energy in bulk ZnO is minimal (60–80 meV), which is in good agreement with experimental data, and indicates weak confinement and increased dielectric screening. These trends underscore the importance of dimensionality in the optical and excitonic manipulation of ZnO to optoelectronics.

5. Defect Optical Signatures

New absorption bands into the visible or near-UV, with defect charge states, relaxation, etc. These were found with

simulated absorption spectra of oxygen vacancies. Charged vacancies displaced both transition energies and oscillator strengths. These were consistent with experimentally observed photoluminescence of broad visible emission in ZnO thin films, which were the result of a variety of defect-related radiative processes.

Discussion

The difference in the bandgap of ZnO films is due to a compromise between quantum confinement which widens the gap in thinner films and surface induced states which can add mid-gap states and decrease the effective gap. These surface states can be suppressed by appropriate surface passivation, e.g. oxygenation of the surface or capping layers so that the confinement-induced blue-shift and intense excitonic characteristics would dominate the optical response.

1. Quantum confinement vs. surface chemistry

Thin-film band-gap variations are regulated by a struggle amid quantum confinement (enhancing the gap) and surface-induced conditions (which may cause a decrease in the gap through mid-gap degrees). To design the device, detrimental surface states can be suppressed by surface passivation (e.g. by oxygenation or by other suitable capping layers) and confinement induced blue-shifts and strong excitonic features can be observed.

2. Importance of accurate electronic structure methods

Semilocal DFT performs dismally well in predicting the gap of ZnO due to self-interaction and localization effects on Zn-3d electrons. Many-body GW corrections and hybrid functionalities are required in order to obtain reliable quasiparticle energies. BSE is necessary to model excitonic effects - in particular it is necessary in thin films where electron-hole binding energies are large and dominate the optical response at the gap. To predictive model, the PBE HSE06/G 0 workflow is an effective one.

3. Role of defects and implications for experiments

We propose that oxygen vacancies and zinc interstitials play a significant role in absorption and emission, and generate features that are associated with our model of defects in ZnO thin films that explain experimental visible PL and sub-band-gap absorption. Energetic levels of the defects and their optical activity are sensitively dependent on the film thickness, on the local relaxation and substrate induced screening.

4. Recommendations for experimental validation

- Angular resolved photoemission spectroscopy (ARPES) and scanning tunneling spectroscopy (STS) would be used to measure the behavior of thin films with a controlled thickness to confirm predicted band dispersions and surface states.
- The excitonic peaks can be resolved and binding energies of films of varying thicknesses can be estimated at optical absorption and reflectance with photoluminescence excitation (PLE) at various temperatures.
- Photoluminescence Photoluminescence with controlled annealing/oxygenation Photoluminescence can be used to correlate defect concentration to sub-band-gap optical properties.

Conclusion

This Research gives a quantum-mechanical analysis of the structural, electronic and optical properties of ZnO thin films, and it is found that their characteristics change with decreasing thickness, surface termination and native defects. The findings verify that ultrathin ZnO films can have strong quantum-confinement effects with resultant widening of the band gaps, high exciton binding energies and the blue shift in the optical absorption edge. Relaxation of the structure at polar and non-polar surfaces causes quantifiable changes in the bond length between Zn and O which directly affect the position of band-edges and optical transition energies. Greater levels of electronic-structure computations including HSE06, GW and BSE were necessary to precisely model quasiparticle energies and excitonic characteristics and in low-dimensional ZnO where screening is weak. The optical spectra revealed that excitonic peaks grow more

intense in films of smaller thickness, whereas native defects, especially oxygen vacancies and zinc interstitials, add sub-band-gap absorption structures to that which is observed to cause visible emission in the experiment. All in all, the results provide definite theoretical standards of interpreting experimental values and determine the rational design of ZnO-based nanoscale optoelectronic and photonic devices. This work has shown important directions to control optical and electronic properties of ZnO thin films by means of confinement, surface chemistry and defect states, which will be employed in the future to tune the technological performances of these materials.

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